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Effects of the sintering atmosphere on the $BaZn_{1/3}Ta_{2/3}O_3$ based Cu multilayer ceramic capacitors

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Abstract

Recent papers report that $BaZn_{1/3}Ta_{2/3}O_3$ (BZT) ceramic can be sintered at a temperature as low as 1050 °C owing to the use of flux agents like B_2O_3 + LiF combined with a slight non-stoichiometry, whereas its usual sintering temperature is 1400 °C. This low sintering temperature (below the Cu's melting point = 1083 °C) opens the route to fabricate copper based multilayer ceramic capacitors, in condition that a reductive atmosphere is used during the sintering. This paper presents the effect of three various sintering atmospheres (air, H_2 (1%) in N_2 and H_2 (1%) in Ar) on the stability and the dielectric properties of BZT. It is researched a suitable sintering atmosphere to prevent Cu from oxidation and to preserve the dielectric properties of BZT. Using the appropriate atmosphere, copper based multilayer ceramic capacitors, with attractive dielectric properties, have been successfully processed.

Keywords: Dielectric properties; Capacitors; Perovskites

1. Introduction

The dielectric $Ba(Zn_{1/3}Ta_{2/3})O_3$ (BZT) is a perovskite type material which exhibits attractive dielectric properties namely, a low losses factor ($\tan(\delta) < 10^{-3}$), a relatively high permittivity (ε_r around 30) and a low temperature coefficient of the permittivity at high frequencies (from MHz to GHz) ($|\tau_\varepsilon| < 100$ ppm/°C[1–4]). These properties make it attractive for fabricating type I multilayer chip capacitors. Nevertheless, BZT ceramic requires high temperature to be correctly sintered (1400 °C), forbidding the use of base metals as electrodes, viz. copper (melting point, m.p. = 1083 °C) and silver (m.p. = 961 °C). Recent papers have reported that BZT ceramic can be sintered at a temperature as low as 1050 °C owing to the use of B_2O_3 + LiF addition combined with a slight non-stoichiometry.[2,3] This result permits the co-sintering between the copper electrodes and the ceramic, providing the stability of the ceramic material during the high temperature treatment. This could open the route to fabricate Cu based multilayer ceramic capacitors. This type of components should be cheaper and more powerful than those co-sintered at high temperatures (>1400 °C) with expensive metals namely Pd, Pt. The aim of this work is to study the behavior of the pure BZT and the 'BZT + additives', for which sintering can be performed at 1050 °C, as a function of the employed atmosphere.

2. Experimental procedure

$Ba(Zn_{1/3}Ta_{2/3})O_3$ and $Ba_{0.99}(Zn_{1/3}Ta_{2/3})O_{2.99}$ powders have been prepared by solid state reaction between $BaCO_3$ (Diopma 99.99%), ZnO (Cerac 99.995%) and Ta_2O_5 (HC Starck 99.9%). All conditions for the synthesis are summarized in the reference 3; in the same manner, the justification to synthesize the $Ba_{0.99}(Zn_{1/3}Ta_{2/3})O_{2.99}$ compound is also given in reference 3 as well as the powder characterization. It can be mentioned that the non-stoichiometry compound has a similar XRD pattern to that of the stoichiometric compound.[3] After the first thermal cycle, the powders have been reground for 45 min in an agate mortar using a planetary grinder in order to achieve a grain size of around one micrometer. For the non-stoichiometric compound, LiF (Prolabo 99%) and B_2O_3 (Prolabo >99%) have been also added according to the following nominal composition $Ba_{0.99}(Zn_{1/3}Ta_{2/3})O_{2.99}$ + 5 mol.% LiF + 10 mol.%. B_2O_3. This formulation named BZT_{LT}, can be sintered at 1050 °C according to our previous study.[3] Eight mm diameter and one mm thickness disks have been shaped using uniaxial pressing at 29 kN.

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Before pressing, an organic binder (polyvinyl alcohol) has been systematically added at around 3 wt% to improve the mechanical behavior of the green samples. Samples have been sintered in three various atmospheres. First, static air has been used as an oxidant atmosphere. Second, a flowing gas of H_2 (1%) in Ar moisture saturated and a flowing gas of H_2 (1%) in N_2 moisture saturated have been used as foaming atmosphere. Moisture saturated atmosphere is used to consume the CO_2 traces and P_{O_2} has been estimated using a zirconia sensor (Setnag). The sintering temperatures were 1400 and 1050° for, respectively, BZT and BZT_{LT}. The sintering time was systematically fixed at 2 h. Dielectrics properties (ε, tan(δ)) were measured at 1 MHz versus temperature (−60 °C/+180 °C) using a LCR bridge (Fluke PM6306) on disks of which each face has been previously covered by an In/Ga eutectic mixture to act as electrodes. Insulating resistivity has been measured using a picoamperemeter (Sefelec). Chemical and microstructural investigations of our samples have been carried out, respectively, by X-ray diffraction (Philips X'Pert, Cu Kα) and SEM observation (SEM Philips XL'30). Copper Based Multilayer ceramics capacitors prototypes have been fabricated by the TEMEX Company and their properties have been carefully examined.

3. Results and discussion

Table 1 summarizes the dielectric properties of the pellets depending of the various processing conditions. It is also indicated the oxygen partial pressure versus the atmosphere used.

3.1. Sintering in air

The first important point is that materials sintered in air (BZT or BZT_{LT}) exhibit the expected properties: density higher than 90% of the theoretical, a relative dielectric constant around 30, a low losses factor (<10^{-3}) and an insulating resistivity higher than 10^{11} Ω cm. Nevertheless, the compound sintered at low temperature (BZT_{LT}) has a slightly increased temperature coefficient (∼103 ppm/°C) compared to BZT (∼0 ppm/°C). The weight loss during sintering is about 3–3.5% which corresponds to the organic binder departure. The good stability of both compounds, when sintering is performed in air, has been obviously expected.

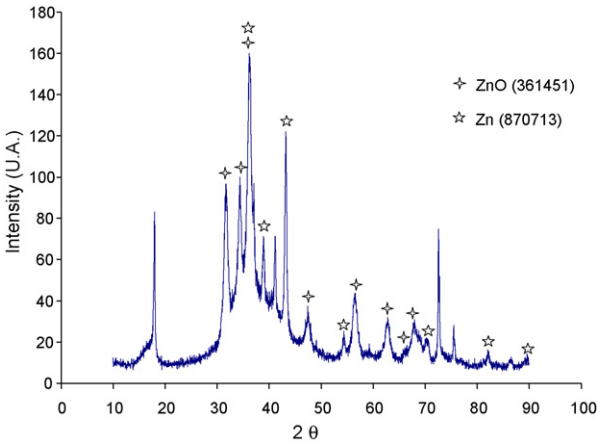

Fig. 1. XRD pattern of the powder which has been found at the cold zone of the furnace.

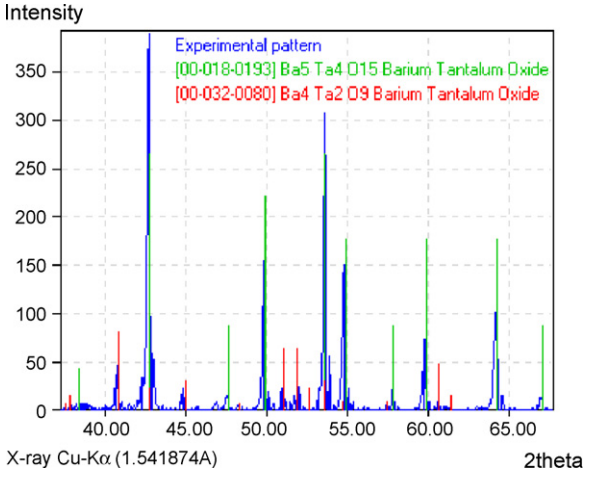

Fig. 2. XRD pattern of the BZT pellet sintered in Ar/H_2 reductive atmosphere.

3.2. Sintering of BZT in low oxygen partial pressure

A very high weight loss of nearly 10% during sintering in N_2/H_2 or Ar/H_2 is observed for BZT (Table 1). During sintering, a black powder has been deposited at the cold zone of the furnace (actually, at the extremities). XRD pattern of this powder (Fig. 1) has allowed to identify it as a mixture of zinc oxide (ZnO number PDF 361451) and metallic zinc (Zn PDF870713). Disks are finally not dense and are mainly composed by $Ba_5Ta_4O_{15}$

Table 1
Properties of the ceramics vs. the processing conditions

	Sintering conditions					Dielectrics properties (RT)			
	T (°C)	Atmosphere	P_{O_2}	Density (%)	$\Delta m/m$ (%)	ε_r	tan(δ)	τ_ε (ppm/°C)	log ρ_i (Ω cm)
BZT	1400	Air	2×10^{-1}	92.5	−3.5	31	$<10^{-3}$	0	15
		$N_2 + 1\%$ H_2	10^{-10}	58	−9.5	24	10^{-2}	162	9
		$Ar + 1\%$ H_2	$\ll 10^{-10}$	57	−9.7	19	$<10^{-3}$	+140	9
BZT_{LT}	1050	Air	2×10^{-1}	90	−3	29	$<10^{-3}$	103	11.7
		$N_2 + 1\%$ H_2	10^{-10}	90	−5	27	$<10^{-3}$	−40	13
		$Ar + 1\%$ H_2	$\ll 10^{-10}$	78	−9	20	$<10^{-3}$	−10	10.2

P_{O_2} is the partial pressure of oxygen; density is the apparent one, $\Delta m/m$ corresponds to the relative weight loss during sintering.

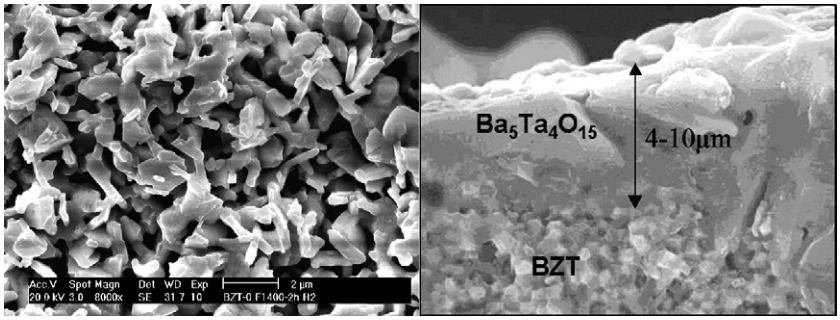

Fig. 3. Microstructure of BZT sintered in Ar/H_2 (left) and microstructure of a section of BZT sintered in pure argon (right).

(PDF00-018-01893) and $Ba_4Ta_2O_9$ (PDF00-032-0080) (Fig. 2). It is remarkable that BZT phase has totally disappeared. In considering these experimental results, it is possible to describe the BZT decomposition mechanism which occurs in very low oxygen partial pressure:

$$9BaZn_{1/3}Ta_{2/3}O_3 \rightarrow Ba_5Ta_4O_{15} + Ba_4Ta_2O_9 + 3ZnO$$

The Fig. 3A shows the microstructure of 'BZT' sintered in Ar/H_2 reductive atmosphere. As it was predicted by the low apparent density, microstructure is very porous and acicular grains characteristics of $Ba_5Ta_4O_{15}$ and $Ba_4Ta_2O_9$ are clearly seen. One additional experiment has been done in pure argon ($P_{O_2} = 5 \times 10^{-6}$ atm). In this case, ZnO departure has been also evidenced as it is testified by the SEM observation of the pellet's cross section (Fig. 3B). It is observed a layer of around 4–10 μm in thickness where $Ba_5Ta_4O_{15}$ is the majority phase. This ZnO loss has been already reported in air when sintering is performed at high temperature.[5] The decomposition of BZT at high temperature and in high reductive atmosphere leads to get very poor dielectric characteristics. Low insulating resistivity (10^9 Ω cm), high losses factor (10^{-2} to 10^{-3}) combined with a lowered permittivity ($\varepsilon_r < 24$) are obtained. The low permittivity could be explained by the high porosity (ε_r (air) = 1) and the presence of other phases, such as $Ba_5Ta_4O_{15}$ which has a permittivity around 28.[6] To summarize, the ZnO departure from BZT is exacerbated by the reductive atmosphere; the lower the oxygen partial pressure, the higher the ZnO volatilization. In extreme condition ($P_{O_2} < 10^{-10}$ atm), BZT phase totally decomposes at 1400 °C. Nevertheless, it is expected that using sintering aids, the sintering temperature lowering should decrease the ZnO departure from the sample.

3.3. Sintering of BZT_{LT} in low oxygen partial pressure

In very aggressive atmosphere (Ar/H_2), disks are not well sintered (density around 70% of theoretical) that leads a decreasing of the permittivity ($\varepsilon_r = 20$). Resistivity is also degraded (10^{10} Ω cm) and XRD analyses show that BZT phase has still disappeared during the sintering. The high weight loss (~9%) accounts for this phenomenon. In N_2/H_2, the density of the sintered disks becomes higher than 90%, the weight loss achieves 5% and the dielectric properties are now proper (Table 1). Nevertheless, SEM observation of the sample shows a porous zone of around 50 μm in deep at the sample periphery (Fig. 4).

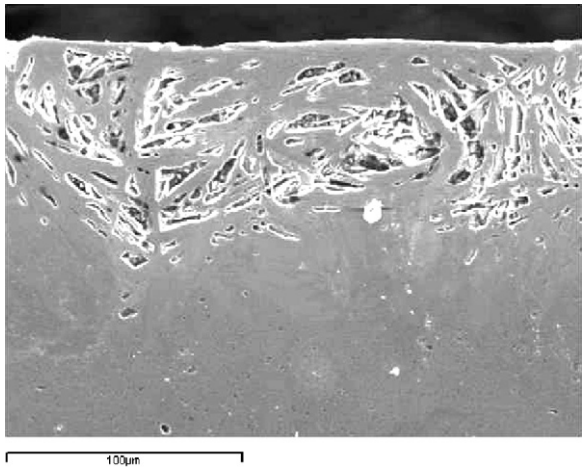

Fig. 4. SEM microstructure of a section of BZT_{LT} sintered in N_2/H_2.

Here again, the bulk is composed by BZT whereas the surface is mainly composed by $Ba_5Ta_4O_{15}$. The good stability of the sample bulk allows the fabrication of copper based multilayer capacitors; this was done by the TEMEX-CERAMICS Company.

3.4. Fabrication of copper based multilayer ceramic capacitors using BZT_{LT}

Five capacitances layers components have been fabricated with copper electrodes; the details of the fabrication are given in references.[1,3] Briefly, all thermal cycles for the debinding and the co-sintering have been performed in N_2/H_2 and the sintering temperature was 1050 °C. Fig. 5 shows the cross section of a typical sintered chip. Ceramics as well as the inner electrodes are

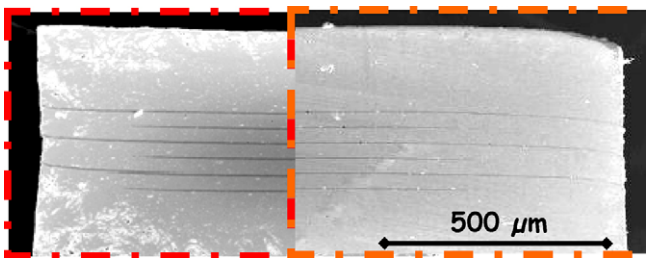

Fig. 5. Cross section of a chip sintered at 1050 °C in N_2/H_2 atmosphere (left). Same component but encapsulated with ZnO powder during sintering (right).

clearly seen. As expected, the periphery of the chip is porous as the result of the partial departure of ZnO. Although this microstructure defect does not affect the dielectric properties, some components have been sintered in embedding the components by a ZnO powder. In this case, it is clearly observed that the porous zone has been totally avoided (Fig. 5). Final dielectric properties of the chips are $C = 24$ pF, $\tan(\delta) < 0.001$, $R_i > 10^{12}\ \Omega$, $\tau_\varepsilon \sim 30$ ppm/°C, values which are similar to those of conventional Pd or Pt based multilayer capacitors.

4. Conclusion

The influence of various reductive atmospheres on the behaviour of two BZT based ceramics has been carried out. The first composition consists in pure BZT which requires 1400 °C to be sintered; the second compound has the following nominal composition $Ba_{0.99}(Zn_{1/3}Ta_{2/3})O_{2.99}$ + 5 mol.% of LiF + 10 mol.% of B_2O_3 and could be sintered at 1050 °C. It has been demonstrated that in Ar/H_2 or in N_2/H_2, the sintering of the pure BZT compound at 1400 °C leads to a total decomposition of the phase in $Ba_5Ta_4O_{15}$ and $Ba_4Ta_2O_9$ due to the ZnO departure. The BZT_{LT} compound sintered in Ar/H_2 at 1050 °C is also totally decomposed. With a less reductive atmosphere (N_2/H_2), the BZT_{LT} compound sintered at the same temperature is also subjected to a ZnO loss but in a lower extend. The BZT phase is stable and only a fine layer of around 50 μm deep at the periphery is subjected to the ZnO loss. Copper based multilayer capacitors using the ceramic BZT_{LT} have been successfully fabricated using N_2/H_2 atmosphere as sintering atmosphere. Such components exhibit very attractive properties, i.e., $C = 24$ pF, $\tan(\delta) < 0.001$, $R_i > 10^{12}\ \Omega$, $\tau_\varepsilon \sim 30$ ppm/°C, showing the high potential of this formulation to fabricate cheaper and high-performance capacitors.

Acknowledgments

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